

# High- $q$ -resolution neutron scattering technique using triple-axis spectrometers

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A new technique is presented that gives a substantial increase in the wavevector  $q$  resolution of triple-axis spectrometers by matching the measurement wavevector  $q$  to the reflection  $\tau_a$  of a perfect-crystal analyzer. A relative Bragg width of  $\delta q/Q \sim 10^{-4}$  can be achieved with reasonable collimation settings. This technique is very useful in measuring small structural changes and line broadenings that cannot be accurately measured with conventional set-ups, while keeping all the strengths of a triple-axis spectrometer.

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## 1. Introduction

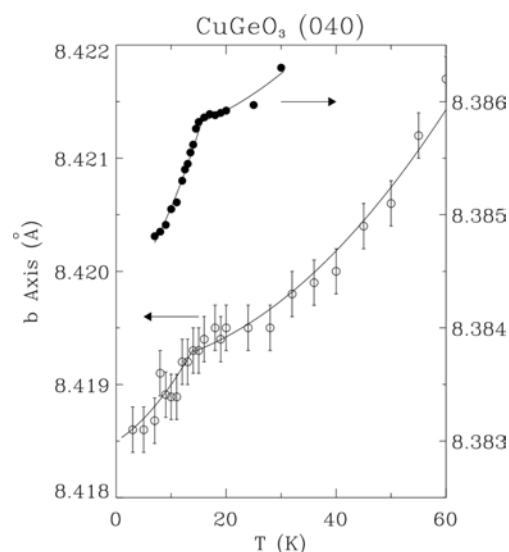
Triple-axis spectrometers (TAS) are widely used in both elastic and inelastic neutron scattering measurements to study the structures and dynamics in condensed matter. They have the flexibility to allow one to probe nearly all coordinates in energy ( $\hbar\omega$ ) and momentum ( $q$ ) space in a controlled manner, and the data can be easily interpreted (Bacon, 1975; Shirane *et al.*, 2002).

The resolution of a triple-axis spectrometer is determined by many factors, including the incident ( $E_i$ ) and final ( $E_f$ ) neutron energies, the wavevector transfer  $\mathbf{Q}$ , the monochromator and analyzer mosaicity, and the beam collimations *etc.* This has been studied in detail (Cooper & Nathans, 1967; Werner & Pynn, 1971; Chesser & Axe, 1973; Popovici, 1975; Popovici *et al.*, 1987). It has long been known that when the measurement wavevector is close to that of the monochromator ( $\tau_m = 2\pi/d_m$ ), good  $q$  resolution can be achieved. This is called the ‘focusing condition’. However, since there is very little freedom to vary the monochromator  $d$  spacing in an experiment, it is usually not possible to achieve focusing near the wavevector of interest.

Recently experiments and calculations have shown that, by matching the reflection of the analyzer ( $\tau_a = 2\pi/d_a$ ) with the measured wavevector, a similar focusing condition can be achieved. The improvement in  $q$  resolution is particularly great when the analyzer is a perfect crystal with very fine mosaicity. However, this is not the case in a conventional triple-axis spectrometer where the monochromator and analyzer crystals are deliberately distorted so that their mosaicity (typically  $\sim 30'$  to  $1^\circ$ ) matches the beam collimations (typically 10 to  $100'$ ). In most cases, using a perfect crystal as analyzer/monochromator only results in much lower intensity because of the large primary extinction and will not improve the resolution significantly. This is because the much coarser beam collimations control the resolution. In the case of elastic scattering measurements, however, the intensity is generally

not a major concern. The Bragg intensity from a reasonable size single crystal (a few grams) can easily reach  $10^4$  counts  $s^{-1}$ . It is therefore feasible to trade off intensity for higher instrumental resolution if the analyzer-side focusing condition can be satisfied.

Lorenzo *et al.* (1994) used this new technique to measure the lattice parameter of  $\text{CuGeO}_3$ . The reciprocal-lattice vector associated with the (040) Bragg peak of  $\text{CuGeO}_3$  matches nearly perfectly with the Ge (220) reflection ( $\tau_{\text{Ge}(220)} = 3.1414 \text{ \AA}^{-1}$ ). The instrument set-up employed a PG(002) monochromator tuned to an incident neutron energy  $E_i = 7.48$  meV, a perfect-crystal Ge (220) analyzer and beam collimations of  $10'-10'-10'-10'$ . A  $q$  resolution of  $0.002 \text{ \AA}^{-1}$

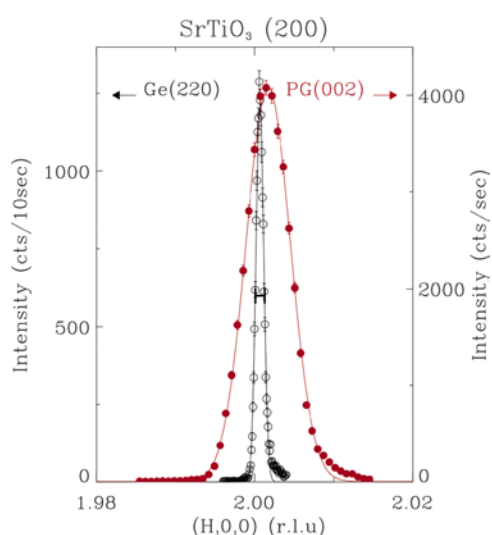


**Figure 1**  
 Temperature dependence of the  $b$  lattice constant of  $\text{CuGeO}_3$  measured at  $\mathbf{Q} = (0, 4, 0)$ , using high- $q$ -resolution neutron scattering [Ge(220) analyzer,  $E_i = 7.48$  meV,  $10'-10'-10'-10'$  collimations, open circles (Lorenzo *et al.*, 1994)] and X-ray diffraction [solid circles (Harris *et al.*, 1994)].

was achieved with this set-up. Fig. 1 compares the lattice parameter of  $\text{CuGeO}_3$  measured by Lorenzo *et al.* (1994) with that found with high-resolution X-ray diffraction measurements of Harris *et al.* (1994). The temperature dependences of these results agree perfectly (lattice spacings offset by  $0.04 \text{ \AA}$ ), demonstrating the potential of this new technique in measuring lattice parameters to a relative accuracy of  $\sim 10^{-4}$ , which is approaching that of high-resolution X-ray diffraction measurements.

This use of the (220) reflection of a perfect Ge crystal analyzer has been further exploited by Ohwada *et al.* (2001), Stock *et al.* (2004) and Gehring *et al.* (2004) in measuring the structure of  $\text{PbXO}_3$ -type relaxor ferroelectrics. These are perovskites with lattice parameter  $a \approx 4.0 \text{ \AA}$ . The length of the reciprocal-lattice vector  $q_{200} \approx 3.14 \text{ \AA}^{-1}$  in these systems is very close to  $\tau_{\text{Ge}(220)}$ . The longitudinal Bragg full width at half-maximum (FWHM) in these measurements is about  $0.003 \text{ \AA}^{-1}$ . Many other ferroelectric perovskite systems, such as  $\text{SrTiO}_3$ , have very similar lattice parameters and thus can be studied with this technique.

One example is shown in Fig. 2, where longitudinal scans along the (200) reflection of single-crystal  $\text{SrTiO}_3$  are plotted. The data were taken on the BT9 triple-axis spectrometer located at the NIST Center of Neutron Research (NCNR). The monochromator is a PG(002) single crystal with a mosaicity of  $\sim 35'$  in both horizontal and vertical directions. The incident neutron energy was  $E_i = 14.7 \text{ meV}$ . The open circles represent data taken using a perfect Ge(220) crystal as analyzer and reasonable beam collimations of  $10'-40'-20'-40'$ . The Bragg peak is extremely sharp, with FWHM  $\delta q/Q \approx 6 \times 10^{-4}$ , almost one order of magnitude better than that obtained using a PG(002) analyzer of mosaicity  $\sim 35'$ , and the finest collimations available ( $10'-10'-10'-10'$ , solid circles in Fig. 2).



**Figure 2**  
A single-crystal  $\text{SrTiO}_3$  longitudinal scan measured at (200) with a PG(002) analyzer,  $10'-10'-10'-10'$  collimations (solid circles), and a Ge(220) perfect-crystal analyzer,  $10'-40'-20'-40'$  collimations (open circles). The resolution for the Ge(220) set-up is represented by the horizontal bar.

In this paper, we describe this new technique in detail using a perfect Ge crystal (mosaicity  $\leq 1'$ ) as analyzer. When the analyzer-side focusing condition is satisfied, and with a proper choice of collimations, an improvement of more than one order of magnitude can be obtained in the longitudinal Bragg width [compared to PG(002) (mosaicity  $\sim 35'$ ) analyzer] with reasonable intensities. By using different reflections of the perfect-crystal analyzer, different Bragg peaks of the sample crystal can be studied. With this technique, we can achieve a reasonably good  $q$  resolution, while retaining the strength and flexibility of a triple-axis spectrometer.

## 2. Resolution calculations

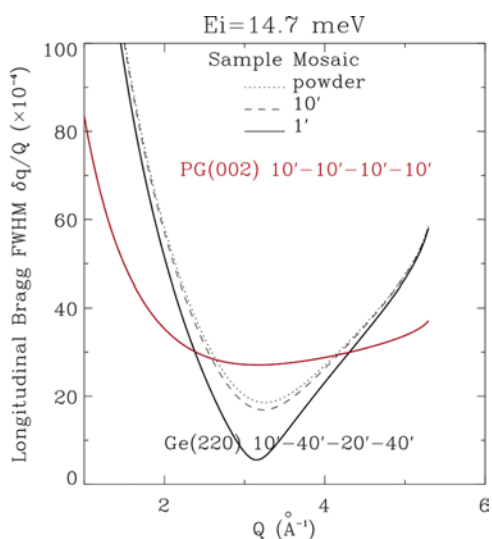
When measuring lattice distortions and structural phase transitions, the magnitude of the Bragg peak splitting, peak width broadening and other related effects are usually proportional to the length of the wavevector. Therefore, the relative Bragg width  $\delta q/Q$  is of more importance than the absolute Bragg width  $\delta q$  itself. In this section, the relative Bragg width  $\delta q/Q$  is calculated for different instrument set-ups and effective sample mosaicities. The calculations are based on the formulas derived by Cooper & Nathans (1967), Werner & Pynn (1971) and Chesser & Axe (1973). The typical instrument set-up of the BT9 triple-axis spectrometer at NCNR was chosen for the calculation/simulations. The monochromator is a vertically focusing PG(002) crystal with mosaicity of  $\sim 35'$ .

With the new technique, a sharp longitudinal Bragg resolution can only be achieved around a small range of  $Q$  that matches the perfect-crystal analyzer reciprocal-lattice vector  $\tau_a = 2\pi/d_a$ . It is much easier to satisfy the focusing condition on the analyzer side than on the monochromator side because the analyzer of a conventional TAS can easily be changed.

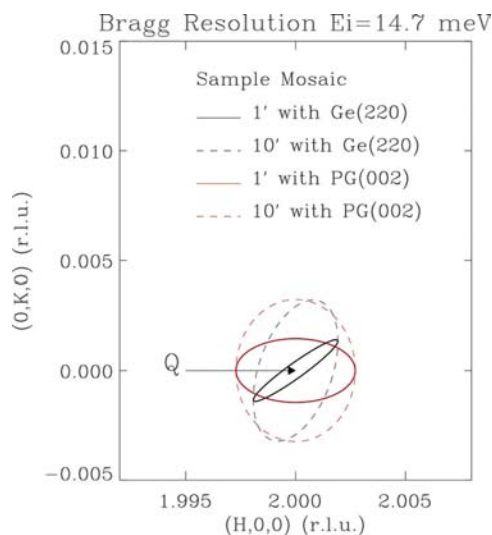
We will first focus our discussion on one specific perfect-crystal analyzer set-up, the Ge(220) reflection, which can be directly compared to the PG(002) analyzer, mostly because their focusing conditions are similar. A detailed comparison is given in Fig. 3. Here we have used beam collimations of  $10'-40'-20'-40'$  and an incident energy of  $14.7 \text{ meV}$  for the Ge(220) set-up. For the PG(002) set-up, we used the best available beam collimation  $10'-10'-10'-10'$  in the calculations. If the sample itself is a perfect single crystal (mosaicity  $\leq 1'$ ), the relative longitudinal Bragg width using a PG(002) analyzer is about  $\delta q/Q \sim 10^{-3}$  to  $10^{-2}$  in the range of  $1 < Q < 6 \text{ \AA}^{-1}$ , with a minimum of  $\sim 2.7 \times 10^{-3}$ . By switching to the perfect Ge crystal (220) [ $\tau_{\text{Ge}(220)} = 3.14131 \text{ \AA}^{-1}$ ] reflection as the analyzer, the longitudinal Bragg width near  $Q = 3.14 \text{ \AA}^{-1}$  is improved by one order of magnitude,  $\delta q/Q \approx 5.5 \times 10^{-4}$ . This value is approaching the Bragg resolution of high-energy X-ray scattering measurements, and about an order of magnitude better than that of a regular neutron powder diffractometer.

Now we consider the effect of non-zero sample mosaicity on the Bragg width. The Bragg widths calculated for an imperfect single-crystal sample (mosaicity  $\eta = 10'$ ) and a powder sample are plotted in Fig. 3 using dashed and dotted lines. The

calculations show that the longitudinal Bragg width of the measurement is greatly affected by the sample mosaicity, but only around those  $q$  values close to  $\tau_a$ . The longitudinal width calculated for a sample crystal with  $10'$  mosaicity is  $\delta q/Q \sim 1.7 \times 10^{-3}$ , more than three times larger than that of a perfect single-crystal sample. A better illustration of the effect of sample mosaicity on the longitudinal resolution is shown in Fig. 4. Here we plot the elastic resolution ellipses around a  $\text{PbXO}_3$  (200) Bragg peak ( $q \approx 3.14 \text{ \AA}^{-1}$ ). For the Ge(220) set-up, when sample mosaicity increases, not only the width but also the shape and orientation of the resolution ellipse changes. This results in a significant increase in the



**Figure 3** Longitudinal Bragg width using a conventional PG(002) analyzer (red lines, beam collimations  $10'-10'-10'-10'$ ) and a Ge(220) perfect-crystal analyzer (black lines, beam collimations  $10'-40'-20'-40'$ ). The dashed and dotted lines are calculations assuming a  $10'$  sample mosaicity and a powder sample.



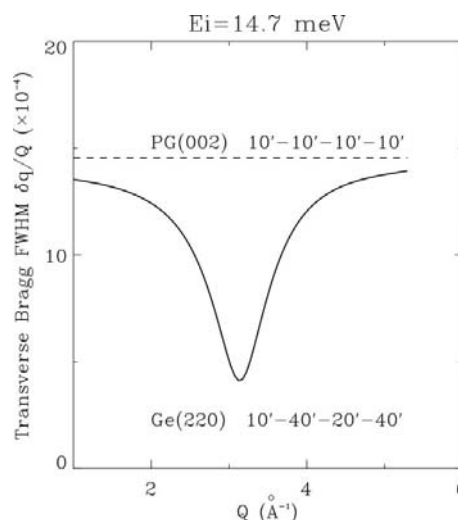
**Figure 4** Bragg resolution ellipse for PG(002) and Ge(220) analyzer set-ups. The beam collimations are  $10'-10'-10'-10'$  for the PG set-up and  $10'-40'-20'-40'$  for the Ge set-up. The dashed lines are results calculated assuming a sample mosaicity  $\eta = 10'$ .

longitudinal Bragg width. Although it is still much better than that of a normal triple-axis set-up, this has to be taken into consideration during the measurements.

If the system being measured undergoes a structural phase transition that causes the crystal effective mosaicity to increase, it will affect measurements in the longitudinal Bragg width as well. In this case, both longitudinal and transverse scans around the Bragg width of interest should be analyzed carefully, with resolution deconvoluted in the analysis in order to obtain the true mosaic/strain information. This is contrary to the case of most synchrotron high-energy X-ray diffraction measurements. In the case of X-ray measurements, the resolution is more defined by the very fine mosaicity of the monochromator and analyzer crystals (usually high-quality Si crystals with mosaicity on the order of  $10^{-3}$  to  $10^{-2}$  arcsec), and the longitudinal Bragg width is unaffected by sample mosaicity.

For wavevectors close to  $q = \tau_a$ , not only longitudinal but also transverse Bragg widths of the measurements are greatly improved. Fig. 5 shows the transverse Bragg width of set-ups using PG(002) and perfect-crystal Ge(220) analyzers. At  $q \approx \tau_a$ , the relative transverse resolution for the set-up with a perfect-crystal Ge(220) analyzer is only limited by the sample mosaicity, yet that of the PG(002) set-up is much broader,  $\delta q_{\perp}/Q \sim 1.5 \times 10^{-3}$  ( $\sim 5'$ ). Therefore, this new technique with perfect Ge crystal analyzer is effective in detecting not only small strains (difference in the length of the Bragg wavevectors) but also small mosaicity distortions and twinings.

In addition to sample mosaicity, another contributing factor to the Bragg resolution is the beam collimation. In Fig. 6, longitudinal Bragg widths using a Ge(220) analyzer with different beam collimations are shown. For  $q$  close to  $\tau_a$ , the width is not affected at all by the beam collimations. This is a natural result since the Bragg width at this point is largely defined by the fine mosaicity of the perfect Ge(220) analyzer



**Figure 5** Transverse Bragg resolution for PG(002) and Ge(220) analyzer set-ups. The collimations are  $10'-10'-10'-10'$  for the PG set-up and  $10'-40'-20'-40'$  for the Ge set-up. The sample mosaicity is chosen to be  $\leq 1'$ .

**Table 1**

Reciprocal spacings  $\tau = 2\pi/d$  for different perfect-crystal analyzers.

PG(002) and (004) are also listed for comparison.

Crystal (reflection)	$\tau$ ( $\text{\AA}^{-1}$ )	SrTiO <sub>3</sub>	
		Reflection	$\tau$ ( $\text{\AA}^{-1}$ )
PG(002)	1.873	(100)	1.609
Ge(111)	1.924		
Si(111)	2.004	(110)	2.276
Cu(111)	3.010	(111)	2.787
Ge(220)	3.141		
Si(220)	3.272	(200)	3.218
Cu(002)	3.476		
Ge(311)	3.683	(210)	3.560
PG(004)	3.747		
Si(311)	3.837	(211)	3.941
Ge(004)	4.442	(220)	4.551
Si(004)	4.628		
Ge(331)	4.841	(300)	4.827
Cu(220)	4.916		
Si(331)	5.043	(310)	5.088

crystal. When  $q$  moves away from this optimum value, one starts to see the effect of resolution broadening by coarser beam collimations. When  $q$  is 10% larger than  $\tau_a$ , the Bragg width can change by a factor of two if the beam collimations change from  $10'-10'-10'-10'$  to  $10'-40'-20'-40'$ . In the course of an experiment, if the wavevector to be measured is very close to  $\tau_a$ , a coarser collimation will help to increase the intensity while not sacrificing too much in resolution. On the other hand, if better  $q$  resolution is essential over a range of  $q$ , then better collimations should be used to obtain better resolution when  $q$  deviates slightly from the focusing condition.

More recently, we noticed that, by using an analyzer reflection with an even larger  $\tau$  value, a better relative Bragg width can be achieved. In Fig. 7, we show the relative longitudinal Bragg widths using different Ge reflections as analyzer crystal, with the same beam collimations  $10'-40'-20'-$

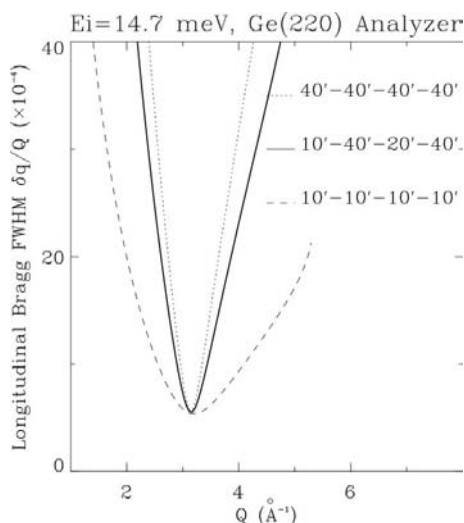
$40'$ . Here we see that, with different analyzer reflections, very good longitudinal Bragg widths ( $\delta q/Q \sim 10^{-4}$ ) can always be achieved when the analyzer-side focusing condition is satisfied. The relative Bragg width improves by almost a factor of two from the set-up using the Ge (220) reflection at  $q \approx 3.14 \text{ \AA}^{-1}$ ,  $\delta q/Q \approx 5.5 \times 10^{-4}$ , to the Ge (004) reflection at  $q \approx 4.44 \text{ \AA}^{-1}$ ,  $\delta q/Q \approx 2.6 \times 10^{-4}$ . More experimental examples will be shown in §4. When changing to reflections with even larger  $\tau_{\text{Ge}}$  values,  $\delta q/Q$  still improves, though not as much.

One can always use other perfect single crystals, *e.g.* perfect Si crystals, perfect Cu crystals *etc.*, as analyzers, to adjust where the minimum of the Bragg width lies in  $q$ , depending on the needs of the experiment. Another choice of perfect-crystal analyzer is SrTiO<sub>3</sub> (see Table 1 for different perfect-crystal analyzers). The advantage of SrTiO<sub>3</sub> is that it has a perovskite structure and all its reflections are available as analyzer reflections, which provide a wider choice of  $\tau$ . A second perfect crystal of the same compound being studied can also be used as the analyzer. Such a choice makes it possible to satisfy the focusing condition at every Bragg reflection of interest.

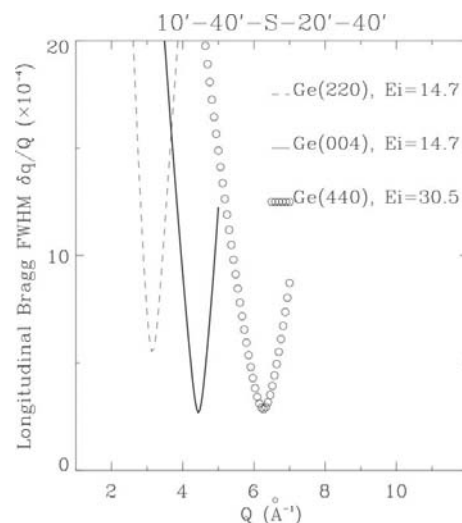
### 3. Intensity simulations

Table 2 shows results from simulations using the *MCSTAS* Monte Carlo code. We have performed simulations for different collimations using the Ge (220) and (004) reflections as analyzers. The relative intensities are calculated at the  $q$  values satisfying the focusing condition, *i.e.*  $q = 4.44 \text{ \AA}^{-1}$  for the Ge (004) set-up and  $q = 3.14 \text{ \AA}^{-1}$  for the Ge (220) set-up.

We note that different sample mosaicities can lead to different relative intensities. With a normal ( $1^\circ$  mosaicity) single-crystal sample, the calculated intensity using a Ge(220) or Ge(004) perfect-crystal analyzer with beam collimations of  $10'-40'-20'-40'$  is about two to five times smaller than that



**Figure 6**  
Longitudinal Bragg width using a Ge(220) perfect crystal analyzer, with different beam collimations.



**Figure 7**  
Longitudinal Bragg width with different analyzers and/or different incident neutron energies  $E_i$ .

**Table 2**

Monte Carlo simulation results on relative intensities of various set-ups with different analyzer reflections and beam collimations.

The incident energy is tuned to 14.7 meV. The monochromator is a PG(002) crystal with mosaicity of  $\sim 35'$ . The intensities are calculated assuming sample mosaicities of 1 and  $60'$ .

Analyzer Collimations	Relative intensity			
	Sample mosaicity $\eta = 60'$		Sample mosaicity $\eta = 1'$	
	$Q = 3.14$ ( $\text{\AA}^{-1}$ )	$Q = 4.44$ ( $\text{\AA}^{-1}$ )	$Q = 3.14$ ( $\text{\AA}^{-1}$ )	$Q = 4.44$ ( $\text{\AA}^{-1}$ )
PG(002) 40'-40'-40'-40'	941	455	73	31
PG(002) 10'-40'-20'-40'			52	18
PG(002) 10'-10'-10'-10'	100	50	23	9
Ge(004) 40'-40'-40'-40'				20
Ge(004) 10'-40'-20'-40'		15		12
Ge(004) 10'-10'-10'-10'				5
Ge(220) 40'-40'-40'-40'			45	
Ge(220) 10'-40'-20'-40'	41		33	
Ge(220) 10'-10'-10'-10'			10	

using a PG(002) analyzer with  $10'-10'-10'-10'$  collimations. On the other hand, when the sample crystal has a very fine mosaicity, e.g.  $1'$ , as used in our calculations, the calculated intensities for the perfect Ge analyzer set-up and PG(002) set-up are not very different.

Because of the huge extinction effects present when using a perfect crystal as the analyzer, the analyzer reflectivity cannot be accurately estimated. The extinction from the Ge perfect-crystal analyzer can greatly reduce the intensity, thus making it impossible to compare intensities between different analyzer set-ups directly. Nevertheless, a comparison between different

collimations using the same analyzer reflection can be quite informative.

For example, with the Ge (004) reflection as the analyzer, the intensity around  $q = 4.44 \text{ \AA}^{-1}$  increases by a factor of two going from  $10'-10'-10'-10'$  to  $10'-40'-20'-40'$ . On the other hand, the resolution coarsens with the collimations, but it depends much on the range of  $q$  being measured. Therefore, one needs to carefully consider the trade-off between resolution and intensity gain/loss in order to choose the most appropriate collimation settings.

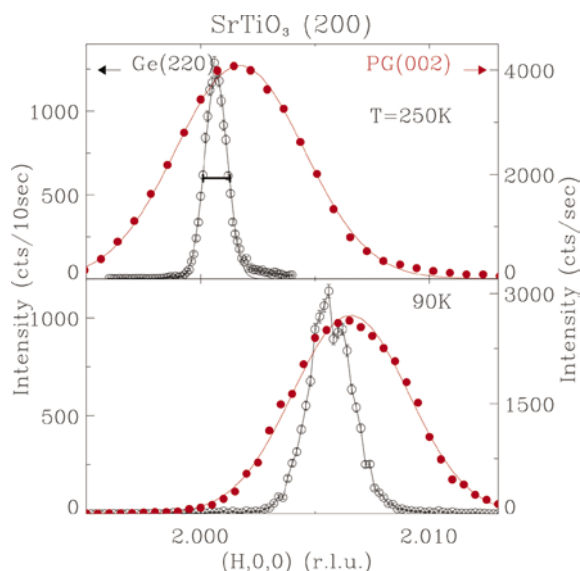
### 4. Examples

In this section, we provide some examples of using this powerful new technique in our measurements.

Fig. 8 shows our measurements of the (200) Bragg peak of an SrTiO<sub>3</sub> single crystal, using a PG(002) monochromator, tuned to  $E_i = 14.7 \text{ meV}$ . Results using a perfect-crystal Ge(220) analyzer (beam collimations of  $10'-40'-20'-40'$ ) and PG(002) analyzer (collimations  $10'-10'-10'-10'$ ) are compared. The measurements are performed on the BT9 triple-axis spectrometer located at the NCNR.

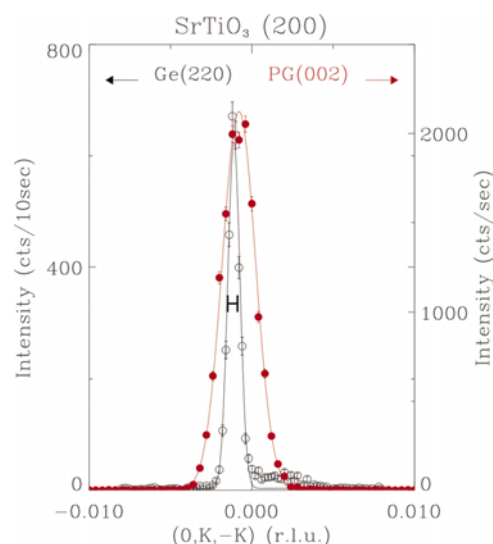
At  $T = 250 \text{ K}$ , the system is in a cubic phase and the Bragg profile is a very sharp Gaussian-shaped peak. With the Ge(220) set-up, the peak width  $\delta q/Q \sim 6 \times 10^{-4}$  is in good agreement with the calculated resolution width ( $\delta q/Q \sim 5.5 \times 10^{-4}$ ).

Below the structural phase transition  $T \sim 110 \text{ K}$ , the system transforms into a tetragonal phase, and a splitting of the (200) Bragg peak occurs. The splitting  $c/a \approx 1.0005$  (Hirota *et al.*, 1995) is very small and almost impossible to measure by conventional neutron triple-axis techniques. We measured the (200) peak with the PG(002) analyzer and the best achievable



**Figure 8**

Longitudinal profiles of the (200) Bragg peaks of SrTiO<sub>3</sub> measured at 250 K (top frame) and 90 K (bottom frame). The horizontal bar indicates the  $q$  resolution along the [200] direction. The incident neutron energy is  $E_i = 14.7 \text{ meV}$ . The collimations are  $10'-40'-20'-40'$  for the Ge(220) set-up (open circles) and  $10'-10'-10'-10'$  for the PG(002) set-up (solid circles).



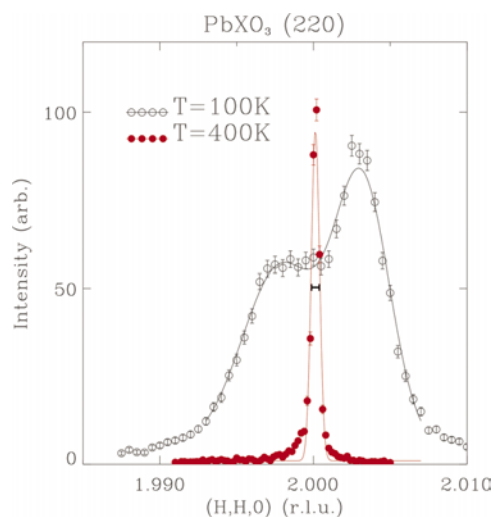
**Figure 9**

Transverse profiles of the (200) Bragg peaks of SrTiO<sub>3</sub> measured at 250 K. The solid lines are fits to Gaussian peaks. The horizontal bar indicates the  $q$  resolution along the [220] direction. The incident neutron energy is  $E_i = 14.7 \text{ meV}$ . The collimations are  $10'-40'-20'-40'$  for the Ge(220) set-up (open circles) and  $10'-10'-10'-10'$  for the PG(002) set-up (solid circles).

beam collimations of  $10'-10'-10'-10'$ , yet no splitting can be detected. With our perfect Ge(220) crystal analyzer set-up, one can clearly see that the original single peak splits into two separate peaks, with  $\Delta q \sim 0.001$  (r.l.u.) ( $\Delta q/Q_{(200)} \sim 5 \times 10^{-4}$ ).

We have also performed transverse scans at the SrTiO<sub>3</sub> (200) Bragg peak, with both the Ge(220) and PG(002) set-ups (Fig. 9). The measurements were done in the cubic phase at  $T = 250$  K. With the Ge(220) analyzer, the scan shows a very sharp peak, with a width  $\delta q/Q \sim 6 \times 10^{-4}$ , slightly broader than the calculated transverse resolution (horizontal bar in the figure)  $\delta q/Q \sim 4.5 \times 10^{-4}$ , indicating a small finite sample mosaicity. Even so, it is still about three to four times sharper than that measured by the  $10'-10'-10'-10'$  collimated PG(002) set-up.

During our recent work on relaxor perovskite PbXO<sub>3</sub> systems, we employed this technique using the (004) reflection of a perfect Ge crystal as the analyzer (Xu *et al.*, 2003; Bai *et al.*, 2004). With larger  $\tau$ , a better relative Bragg width has been achieved. In Fig. 10, we show longitudinal scans at a (220) Bragg peak of PbXO<sub>3</sub> [ $X = (\text{Mg}_{1/3}\text{Nb}_{2/3})_{0.73}\text{Ti}_{0.27}$ ] (Xu *et al.*, 2003). Here,  $q \approx 4.4 \text{ \AA}^{-1}$ , very close to  $\tau_{\text{Ge}(004)} = 4.442 \text{ \AA}^{-1}$ . For  $T$  above the Curie temperature  $T_C \approx 375$  K, the system is in the cubic phase and we have a very sharp (220) peak,  $\delta q/Q \sim 3 \times 10^{-4}$ . With the Ge(004) set-up, we can achieve a relative Bragg width almost half of that using the Ge(220) reflection. For  $T$  below  $T_C$ , the system transforms into a rhombohedral phase, and a splitting of the (220) Bragg peak occurs. The splitting of the two peaks is  $\Delta q/Q_{(220)} \sim 2.5 \times 10^{-3}$ , and the width of the two split peaks is  $\delta q/Q_{(220)} \sim 2.2 \times 10^{-3}$ . With our perfect Ge(004) crystal analyzer set-up, not only the splitting but also the broadening



**Figure 10**  
Profiles of the (220) Bragg peaks for PbXO<sub>3</sub> [ $X = (\text{Mg}_{1/3}\text{Nb}_{2/3})_{0.73}\text{Ti}_{0.27}$ ] measured with a Ge(004) perfect-crystal analyzer, are shown at 100 K (open circles) and 500 K (solid circles). The beam collimations are  $10'-40'-20'-40'$  and  $E_i = 14.7$  meV. The solid lines are fits to Gaussians. The horizontal bar indicates the  $q$  resolution along the [220] direction.

of the two peaks, which is a result of the internal strain, can be accurately measured.

## 5. Summary

Based on the previous calculations and experimental examples, we show that a huge improvement in the relative Bragg width can be achieved when the analyzer-side focusing condition is satisfied by matching the measurement wavevector to a perfect-crystal analyzer reflection. With this new technique, one can still enjoy the flexibility of a triple-axis spectrometer, while improving the moderate relative  $q$  resolution of conventional TAS by one to two orders of magnitude.

Owing to the primary extinction from the perfect analyzer crystal, the Bragg intensity using this new technique is usually a few orders of magnitude smaller than a conventional PG(002) analyzer set-up (see Fig. 2). For elastic measurements, this is usually not the problem if using a reasonable size single crystal. By using different collimations, the intensity/resolution trade-off can be tuned to best fit the desired measurements.

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## References

- Bacon, G. E. (1975). *Neutron Diffraction*. Oxford: Clarendon Press.
- Bai, F., Wang, N., Li, J., Viehland, D., Gehring, P., Xu, G. & Shirane, G. (2004). *J. Appl. Phys.* **96**, 1620–1627.
- Chesser, N. J. & Axe, J. D. (1973). *Acta Cryst.* **A29**, 160–169.
- Cooper, M. J. & Nathans, R. (1967). *Acta Cryst.* **23**, 357–367.
- Gehring, P. M., Chen, W., Ye, Z.-G. & Shirane, G. (2004). *J. Phys. Condens. Matter*, **16**, 7113–7121.
- Harris, Q. J., Feng, Q., Birgeneau, R. J., Hirota, K., Kakurai, K., Lorenzo, J. E., Shirane, G., Hase, M., Uchinokura, K., Kojima, H., Tanaka, I. & Shibuya, Y. (1994). *Phys. Rev. B*, **50**, 12606–12610.
- Hirota, K., Hill, J. P., Shapiro, S. M., Shirane, G. & Fujii, Y. (1995). *Phys. Rev. B*, **52**, 13195–13205.
- Lorenzo, J. E., Hirota, K., Shirane, G., Tranquada, J. M., Hase, M., Uchinokura, K., Kojima, H., Tanaka, I. & Shibuya, Y. (1994). *Phys. Rev. B*, **50**, 1278–1281.
- Ohwada, K., Hirota, K., Rehrig, P. W., Gehring, P. M., Noheda, B., Fujii, Y., Park, S.-E. E. & Shirane, G. (2001). *J. Phys. Soc. Jpn*, **70**, 2778–2783.
- Popovici, M. (1975). *Acta Cryst.* **A31**, 507–513.
- Popovici, M., Stoica, A. D. & Ionita, I. (1987). *J. Appl. Cryst.* **20**, 90–101.
- Shirane, G., Shapiro, S. M. & Tranquada, J. M. (2002). *Neutron Scattering with a Triple-Axis Spectrometer*. Cambridge University Press.
- Stock, C., Birgeneau, R. J., Wakimoto, S., Gardner, J. S., Chen, W., Ye, Z.-G. & Shirane, G. (2004). *Phys. Rev. B*, **69**, 094104–1–10.
- Werner, S. A. & Pynn, R. (1971). *J. Appl. Phys.* **42**, 4736–4749.
- Xu, G., Viehland, D., Li, J. F., Shirane, G. & Gehring, P. M. (2003). *Phys. Rev. B*, **68**, 212410–1–4.